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Anisotropic magnetism, superconductivity, and the phase diagram of $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ V. Tsurkan,^{1,2} J. Deisenhofer,¹ A. Günther,¹ H.-A. Krug von Nidda,¹ S. Widmann,¹ and A. Loidl¹¹*Experimental Physics 5, Center for Electronic Correlations and Magnetism, Institute of Physics, University of Augsburg, DE-86159, Augsburg, Germany*²*Institute of Applied Physics, Academy of Sciences of Moldova, MD-2028, Chisinau, Republic of Moldova*

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We report the crystal growth and structural, magnetic, conductivity, and specific heat investigations of $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ single crystals with varying stoichiometry prepared by self-flux and Bridgman methods. The system exhibits a strongly anisotropic antiferromagnetic behavior below 400 K. Bulk superconductivity is found in samples with Fe concentrations $1.53 < 2 - y < 1.6$, whereas for $2 - y < 1.5$ and $2 - y > 1.6$, insulating and semiconducting behavior is observed, respectively. Within the measured range of variation of the Rb concentration (0.6–0.8) no correlation between the Rb content and the lattice parameters of the samples was found. The superconducting samples show the smallest value of the lattice parameter c compared to the nonsuperconducting samples. The sharpest transition to the superconducting state, the highest transition temperature T_c of 32.4 K, and the highest diamagnetic response corresponding to a critical current density j_c of 1.6×10^4 A/cm² (at 2 K) is found for compositions close to $\text{Rb}_2\text{Fe}_4\text{Se}_5$. Upper critical fields H_{c2} of ~ 250 kOe for the in-plane and 630 kOe for the interplane configurations are estimated from resistivity studies in magnetic fields. In the nonsuperconducting samples with the Fe concentration below 1.45, both specific heat and susceptibility revealed an anomaly at 220 K, which is not related to antiferromagnetic or structural transformations. Comparison with the magnetic behavior of nonsuperconducting samples provides evidence for the coexistence of superconductivity and static antiferromagnetic order.

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I. INTRODUCTION

Iron-based superconductors^{1–4} are currently among the most intensively studied materials. Among different groups of these superconductors, the iron chalcogenides have recently attracted particular attention. The undoped iron selenide, FeSe, exhibits a relatively low critical temperature $T_c \simeq 8$ K at ambient pressure,⁴ but it raises to 37 K under external pressure.^{5,6} Earlier attempts to increase T_c of FeSe using chemical doping resulted in a $T_c \simeq 14$ K by substitution of Se with Te.^{7,8} Recent reports of $T_c \simeq 30$ K in potassium intercalated FeSe⁹ further pushed the research activity in the iron chalcogenide family. Consequently, successful intercalation of other alkali metals (Rb and Cs) in FeSe was realized and superconducting (SC) samples with T_c between 27 and 33 K were prepared.^{10–16} Further studies of SC chalcogenides with hypothetical stoichiometry $\text{A}_{0.8}\text{Fe}_2\text{Se}_2$ ($\text{A} = \text{K}, \text{Rb}, \text{Cs},$ and Tl) revealed significant differences in their SC properties compared to the related SC pnictides with a similar structural arrangement.

Intriguing coexistence of superconductivity and static antiferromagnetic order and a proximity to an insulating state were suggested.^{17–21} In addition, angle-resolved photoemission spectroscopy (ARPES) studies^{22–24} showed a different topology of the Fermi surface in $\text{A}_{0.8}\text{Fe}_2\text{Se}_2$ compared to other Fe-based superconductors suggesting a pairing mechanism distinct from $s\pm$ symmetry.²⁵ Despite numerous reports, the data on $\text{A}_{0.8}\text{Fe}_2\text{Se}_2$ materials with regard to their intrinsic properties is far from being conclusive. For example, the conducting and magnetic properties of the Rb-based samples reported by different authors vary significantly indicating a strong dependence on preparation conditions and impurity content.^{12,13,15,16,19,20} Moreover, the correlation between the properties and stoichiometry of $\text{A}_{0.8}\text{Fe}_2\text{Se}_2$ has been not fully

established yet. Here, we present the results of the structural, magnetic, conductivity, and thermodynamic characterization of the Rb-Fe-Se system performed on single crystals grown by two different methods: self-flux and Bridgman techniques. The selection of the Rb-based system is motivated by an ease to grow large single crystals with a higher volume fraction of the SC phase compared to K- and Cs-based Fe chalcogenides. The variations of the conducting and magnetic parameters with the stoichiometry determined by wavelength dispersive x-ray electron-probe microanalysis (WDS EPMA) are summarized in a phase diagram of the Rb-Fe-Se system.

II. RESULTS AND DISCUSSION**A. Preparation and composition analysis**

Polycrystalline FeSe synthesized from the high-purity elements (99.98% Fe and 99.999% Se) and 99.75% Rb were used as a starting material for the growth of single crystals. Handling of the reaction mixtures was done in an argon box with residual oxygen and water content less than 1 ppm. The preparation conditions are given in Table I. In the growth runs with starting composition corresponding to nominal stoichiometry $\text{Rb}_{0.8}\text{Fe}_2\text{Se}_2$, the solidified ingots showed significant inhomogeneity. Samples from the top of the ingots revealed superconducting properties whereas the bottom of the ingots contained poorly crystallized material with ferromagnetic behavior. A similar separation was observed for the growth runs with the starting stoichiometry $\text{Rb}_{0.9}\text{Fe}_2\text{Se}_2$, although the samples from the top of the ingot were not superconducting. In the growth runs with the starting stoichiometry $\text{Rb}_{0.8}\text{Fe}_{1.6}\text{Se}_2$, such a strong phase separation was not observed, but the samples were also nonsuperconducting.

TABLE I. Preparation conditions, EPMA, and x-ray data for $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples. (Numbers in brackets indicate standard deviations.)

Sample	Preparation conditions				Concentration			Starting mixture	Lattice constant	
	Method	Soaking temperature	Soaking time	Cooling rate	Rb (1-x)	Fe (2-y)	Se		<i>a</i> , <i>b</i> (Å)	<i>c</i> (Å)
F 266	self-flux	1030 °C	3 h	6 °C/h	0.653(30)	1.532(8)	2.000(35)	Rb+2.5FeSe	3.9285(3)	14.6003(18)
F 274	self-flux	1030 °C	3 h	6 °C/h	0.672(15)	1.430(8)	2.000(17)	Rb+2FeSe+0.5Se	3.9270(9)	14.6155(49)
F 286	self-flux	1030 °C	3 h	6 °C/h	0.609(22)	1.608(9)	2.000(33)	Rb+2.5FeSe+0.5Se	3.9180(11)	14.6273(61)
F 295	self-flux	1030 °C	3 h	6 °C/h	0.724(18)	1.560(10)	2.000(19)	Rb+2.5FeSe	3.9310(7)	14.5914(36)
BR 16-1	Bridgman	1070 °C	3 h	3 mm/h	0.796(28)	1.596(8)	2.000(37)	Rb+2.5FeSe	3.9228(7)	14.5909(38)
BR 16-8	0.801(28)	1.602(9)	2.000(37)	...	3.9304(6)	14.6050(28)
BR 16-10	0.759(42)	1.601(9)	2.000(47)
BR 17-1	Bridgman	1070 °C	3 h	3 mm/h	0.737(26)	1.439(8)	2.000(38)	0.9Rb+2FeSe	3.9117(5)	14.6516(36)
BR 17-4	0.733(24)	1.439(9)	2.000(38)
BR 17-5	0.733(24)	1.432(15)	2.000(23)
BR 17-10	0.786(17)	1.459(10)	2.000(15)
BR 18	Bridgman	1070 °C	5 h	3 mm/h	0.659(32)	1.537(9)	2.000(36)	Rb+2.5FeSe	3.9282(4)	14.5899(23)
BR 19	Bridgman	1070 °C	5 h	3 mm/h	0.698(17)	1.497(11)	2.000(25)	Rb+2FeSe+0.5Se	3.9379(1)	14.5983(7)
BR 22	Bridgman	1070 °C	5 h	3 mm/h	0.644(31)	1.620(12)	2.000(35)	Rb+2.5FeSe+0.5Se	3.9021(7)	14.6484(58)
BR 26	Bridgman	1070 °C	5 h	3 mm/h	0.740(36)	1.600(6)	2.000(25)	Rb+2.5FeSe	3.9190(9)	14.5515(57)
BR 28	Bridgman	1070 °C	5 h	3 mm/h	0.685(21)	1.588(9)	2.000(40)	Rb+2.5FeSe	3.9251(6)	14.5932(36)

The deviation from the nominal stoichiometry is a general problem for the intercalated iron chalcogenides. In most earlier reports, the composition of samples was determined by semi-quantitative energy dispersive x-ray (EDX) analysis or more precise ICP analysis. Both methods of compositional analysis revealed a different stoichiometry compared to the nominal $\text{Rb}_{0.8}\text{Fe}_2\text{Se}_2$.^{12,13,15} Various ranges of iron stoichiometry for the superconducting samples were reported, but the accuracy of these methods appears not to be sufficient to detect subtle changes in the composition. Therefore the composition of the grown samples was determined by WDS EPMA that allows to reach an accuracy of 0.5% for Fe and 1% for Se. The concentration of the elements in the samples was measured using a Cameca SX50 analyzer on freshly cleaved samples minimally exposed to air (~ 1 min) to prevent oxidation. The EPMA data are also given in Table I. They represent the values averaged over multiple (at least ten) measured spots with an area of $80 \times 60 \mu\text{m}^2$. The concentrations of Rb and Fe elements are normalized assuming a Se concentration of two per formula unit. The EPMA analysis revealed notable deviations in the composition of the samples from the starting stoichiometry for all prepared batches. The concentration of Fe in the superconducting samples from different batches was found to vary between 1.53 and 1.60, while the variations of the Rb concentration were between 0.65 and 0.8. The Rb deficiency is probably caused by a strong interaction with quartz glass at high temperatures. The stoichiometry of the samples exhibiting the highest SC parameters (from batches BR 16 and BR 26 described below) was close to $\text{Rb}_{0.8}\text{Fe}_{1.6}\text{Se}_2$ (or $\text{Rb}_2\text{Fe}_4\text{Se}_5$ in other notation). The concentration of Fe in the samples prepared from exactly the same nominal mixture $\text{Rb}_2\text{Fe}_4\text{Se}_5$ was, however, less than ~ 1.5 and the samples were not superconducting. Another group of nonsuperconducting samples was also found, with the Fe concentration being higher than ~ 1.6 . The variations of the Fe and Se concentrations on the individual samples were less than 1 and 2%, respectively,

indicating their high homogeneity. The maximal variations of the Rb concentrations were of about 5%, which may indicate an inhomogeneous distribution of Rb ions. However, we do not exclude that this can be also an artifact due to observed sensitivity of the counting rate to a slight misalignment of the sample due to remaining fragments of the removed layers on the surface after cleavage and strong sensitivity of Rb to oxidation.

B. X-ray diffraction and structure

Figures 1 and 2 demonstrate the room-temperature x-ray powder diffraction patterns obtained on crushed single crystals using a STOE Stadi P diffractometer with $\text{Cu K}\alpha$ radiation, $\lambda = 1.54056 \text{ \AA}$ for superconducting and nonsuperconducting samples, respectively. The lattice parameters of the samples were obtained from the Rietveld refinement using the

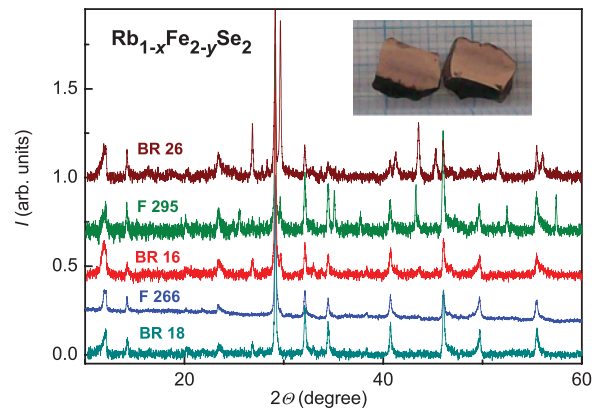


FIG. 1. (Color online) X-ray diffraction patterns for different superconducting $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples. Inset: image of the single crystals BR 26 subtracted from the ingot. Sample labels BR and F stand for those grown by Bridgman and flux methods, respectively.

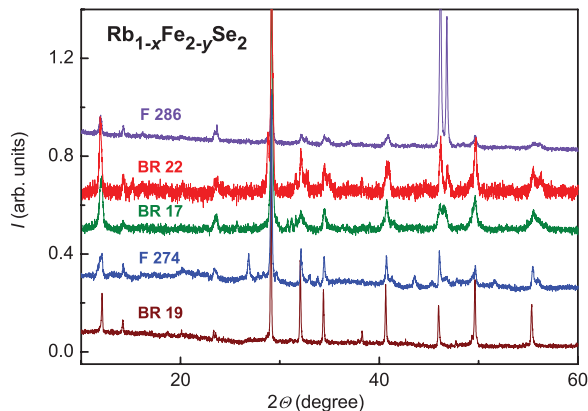


FIG. 2. (Color online) X-ray diffraction patterns for different nonsuperconducting $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples.

FULLPROF SUITE²⁶ assuming tetragonal symmetry $I4/mmm$. They are presented in Table I. The refinement was done using the site occupations for constituent elements as determined by EPMA.

It is important to note that nonsuperconducting samples from the batch BR 19 with Fe concentration $\simeq 1.5$ show the narrowest diffraction lines. X-ray spectra of the nonsuperconducting samples from other batches show additional features, which are probably related to the difference in their vacancy structure. For example, samples with the Fe concentration below 1.45 exhibit much broader peaks indicating inhomogeneity and stress. Samples with the Fe concentration above 1.61 exhibit splitting of peaks, which can be attributed to orthorhombic distortions. We also noticed that superconducting samples exhibit much broader x-ray diffraction peaks and additional reflexes beside the main peaks. In the superconducting samples from batches F 295, BR 16, and BR 26, the additional reflexes are of a considerable magnitude. Additional peaks, although of smaller magnitudes, were also observed in the single-crystal x-ray diffraction spectra in a number of SC $\text{A}_{0.8}\text{Fe}_{1-x}\text{Se}_2$ compounds but at lower angles compared to the main peaks.^{11,13,16,27} These additional peaks were attributed either to inhomogeneous distribution of the alkali ions or to a modulated structure resulting from ordering of iron vacancies. The absence of such peaks in the nonsuperconducting samples probably exclude any significant inhomogeneous distribution of the Rb ions in our samples. Recent single-crystal x-ray and neutron diffraction studies of a number of $\text{A}_{0.8}\text{Fe}_2\text{Se}_2$ compounds identified their crystal structure within a tetragonal $I4/m$ space group symmetry.^{21,28–30} The x-ray patterns of our samples were also refined within the $I4/m$ symmetry. For example, for the SC sample BR 16 the obtained value of the lattice constant $a = 8.7977$ Å agrees well with $\sqrt{5}a$ of the supercell. We must note that the x-ray pattern only of the nonsuperconducting sample BR 19 with the sharpest diffraction peaks is fully compatible with the $I4/m$ space group symmetry. However, within the $I4/m$ symmetry, it was not possible to refine additional peaks observed for the superconducting samples BR 16 and BR 26, suggesting a second crystallographic phase coexisting with the main phase. The coexistence of different crystallographic

phases, was earlier reported for nonsuperconducting $\text{K}_{1-x}\text{Fe}_{2-y}\text{Se}_2$.³¹ However, the orthorhombic symmetry $Pmna$ of the additional phase in Ref. 31 is incompatible with the symmetry of the second crystallographic phase observed in our SC samples. The phase separation in intercalated iron chalcogenides is now actively investigated. Recent transmission electron microscopy,³² x-ray diffraction,³³ ARPES,³⁴ optical spectroscopy,³⁵ and scanning tunneling microscopy studies^{36,37} provided evidence for the phase separation in K-intercalated iron superconductors. Very recently, inelastic neutron scattering,³⁸ optical spectroscopy,³⁹ and Mössbauer studies⁴⁰ of the superconducting $\text{Rb}_{0.8}\text{Fe}_{1.6}\text{Se}_2$ samples (batch BR 16) confirmed the phase separation scenario also for the Rb-based superconductors. However, the relevance to phase separation of the second phase observed in the x-ray spectra of the superconducting samples is still an open question and additional studies are necessary to clarify its origin.

Since the quality of the refinement of the x-ray powder-diffraction data within the $I4/m$ symmetry was lower compared to the $I4/mmm$ symmetry due to reduced statistics and a larger number of refined parameters, we present the data of the refinement within the $I4/mmm$ symmetry. The analysis of the obtained data shows only slight variations of the lattice parameter a with the Fe concentration in the range 1.53–1.6 and a decreasing tendency below 1.5 and above 1.6. An opposite trend is found for the variation of the c parameter. The largest value of the lattice parameter a and the largest value of the unit cell volume are observed for the nonsuperconducting sample BR 19 with Fe concentration $\simeq 1.5$. With exception of this sample, the values of the c parameter are lower for the superconducting samples, compared to the nonsuperconducting ones. Surprisingly, within the measured range of variation of the Rb concentration in the samples (0.6–0.8) no correlation between the Rb concentration and the value of the c parameter was found, although one would intuitively expect an increase of the c parameter with increasing Rb content. This probably can be attributed to a difference in the defect structure of the samples, which requires a more comprehensive study. To solve remaining issues concerning the defect structure, vacancy ordering, and the symmetry of the additional crystallographic phase, single-crystal x-ray investigations are desirable.

Finally, we are also documenting the results of the microstructural study. We observed that samples with the Fe concentration in the range of 1.53–1.6 have very flat surfaces without visible defects or precipitates. They can be easily cleaved along the ab plane reflecting the two-dimensional structure of the crystals. The samples with the Fe content below 1.45 are also rather easily cleaved; their surface contains a large number of dotlike defects. The microstructure of the samples with the Fe concentration above 1.6 is considerably different compared to that of the samples with a lower Fe content. They are mostly three-dimensional and therefore difficult to cleave along the ab plane.

C. Magnetic susceptibility

Figures 3 and 4, respectively, present the magnetic susceptibility versus temperature for superconducting and nonsuperconducting samples from different growth runs measured in a magnetic field $H = 10$ kOe applied parallel to the c axis. The

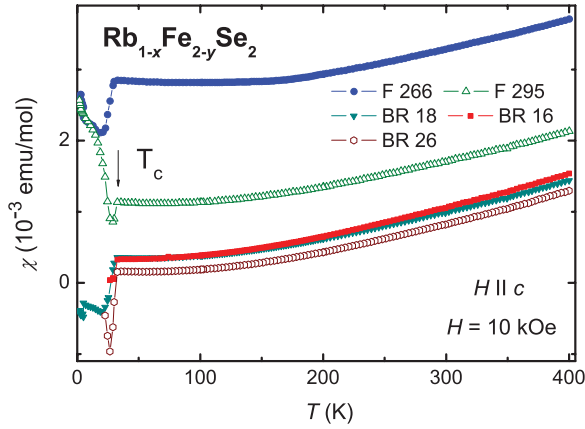


FIG. 3. (Color online) Temperature dependencies of the field-cooled susceptibility for superconducting $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples from different batches measured in a field of 10 kOe applied along the c axis. The arrow indicates the SC transition temperature.

measurements were performed on cooling using a SQUID magnetometer MPMS 5 (Quantum Design). On lowering temperature from 400 to 33 K, the susceptibility for all SC samples continuously decreases. The SC samples prepared by self-flux from the starting composition $\text{Rb}_{0.8}\text{Fe}_2\text{Se}_2$ exhibit a higher susceptibility than those prepared by Bridgman method, which probably can be attributed to the presence of a small amount of nonreacted iron, e.g., of $\sim 0.3\%$ estimated from the magnetic moment for the sample F 266 measured at 40 K, which exhibits saturation at low fields. In contrast, the magnetization of the SC samples grown by Bridgman method shows nonsaturated behavior (see below) and the amount of the nonreacted iron in these samples appears to be significantly reduced. In the SC samples, a clear downturn of χ at the transition temperature at around 30 K is evidenced, while χ for the nonsuperconducting samples shows a small upturn toward lower temperatures. It is important to note the very close values and essentially similar temperature dependence of the susceptibility at temperatures above 33 K both for the

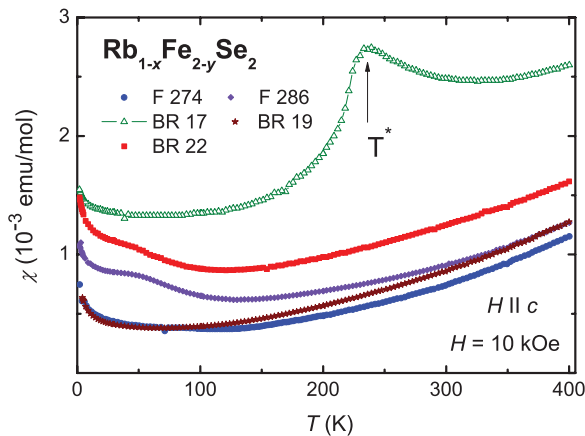


FIG. 4. (Color online) Temperature dependence of the field-cooled susceptibility for nonsuperconducting $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples from different batches measured in a field of 10 kOe applied along the c axis. The arrow marks the temperature of the maximum of the susceptibility T^* for sample BR 17.

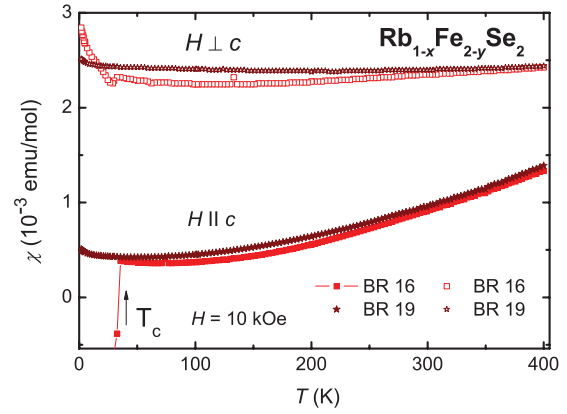


FIG. 5. (Color online) Temperature dependence of the field-cooled susceptibility for two SC (BR 16) and non-SC (BR 19) $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples measured in a field of 10 kOe applied parallel and perpendicular to the c axis.

SC samples from batch BR 16 and non-SC samples from batch BR 19 despite a clear difference in their chemical composition. The susceptibility of the exceptional sample BR 17 (with the strongly reduced Fe concentration below 1.46) shows a nonmonotonic temperature dependence with a broad maximum at around $T^* = 233$ K (for $H||c$), which is also evidenced in the specific heat (see below).

In Fig. 5, the temperature dependence of the magnetic susceptibility is shown for two selected SC and non-SC samples (BR 16 and BR 19) measured in a magnetic field $H = 10$ kOe applied parallel and perpendicular to the c axis. The susceptibility χ_{\perp} for the perpendicular configuration ($H \perp c$) is by a factor of five higher than for $H||c$, and shows a weak temperature dependence, unlike the significant decrease of $\chi_{||}$ on decreasing temperature. Such behavior of the χ_{\perp} and $\chi_{||}$ is characteristic for an anisotropic antiferromagnet with the c axis being the direction of alignment of the spins. A similar anisotropy of the magnetic properties has been recently reported for insulating $\text{TiFe}_{1.6}\text{Se}_2$ ⁴¹ with a Néel temperature of 430 K. The anisotropic antiferromagnetism appears to be a distinct feature of the whole family of the intercalated iron chalcogenides.

In order to determine the magnetic ordering temperature, we measured the susceptibility of the sample BR 16 at temperatures up to 650 K using a heating option of the SQUID. The measurements (not shown) revealed a poorly pronounced maximum in the susceptibility at a temperature of $527(\pm 1)$ K. At the same time, a strongly irreversible behavior of $\chi(T)$ on successive cooling and warming cycles was detected. Subsequent low-temperature measurements of the sample after the high-temperature cycling showed a strongly reduced amount of the SC phase indicating a partial degradation of the sample. Therefore it is unclear whether 527 K represents a true Néel temperature for this sample. Thus, to get an independent estimate of the Néel temperature, we used an extrapolation of the $\chi(T)$ data for the two measured configurations assuming that at T_N they should merge. The data on χ_{\perp} and $\chi_{||}$ above 100 K were fitted by polynomials of second order. From the extrapolation of χ_{\perp} and $\chi_{||}$, a Néel temperature of about $620(\pm 5)$ K was estimated for SC sample BR 16.

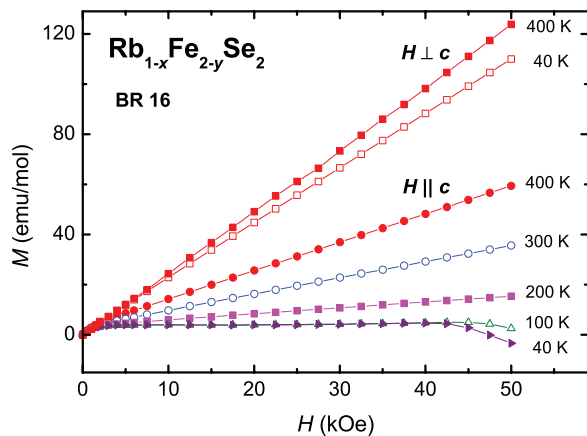


FIG. 6. (Color online) Magnetization curves for the SC sample BR 16 for different temperatures measured with the magnetic field applied along the c axis and within the ab plane.

In Figs. 6 and 7, respectively, the magnetization curves $M = f(H)$ for the superconducting sample BR 16 and nonsuperconducting sample BR 19, measured at different temperatures are presented. The data are shown for the magnetic field applied parallel and perpendicular to the c axis. In the SC sample, M_{\parallel} shows a small nonlinearity at low fields and a linear increase at high fields. M_{\parallel} increases significantly with temperature from 40 to 400 K. At the same time, M_{\perp} being considerably higher than M_{\parallel} , shows a weak temperature dependence in the temperature range 40–400 K in sharp contrast to M_{\parallel} . It is worth noting here that the magnetization of the non-SC sample exhibits a field and temperature dependence for both field configurations very similar to that observed for the SC sample. Importantly, the magnetization of the non-SC sample is not changing in the range from 40 to 2 K as demonstrated by the coinciding data for 2 and 40 K in Fig. 7. A close similarity of the magnetization behavior observed for both the SC and non-SC samples suggests that the static magnetic order coexists with the superconducting state.

Figure 8 shows the temperature dependencies of the zero-field cooled (ZFC), χ_{ZFC} , and field-cooled (FC), χ_{FC} ,

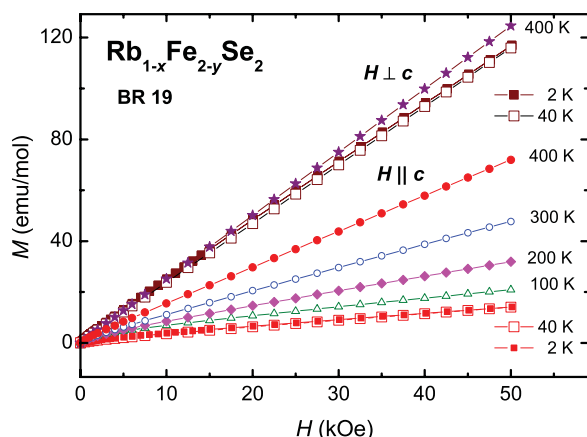


FIG. 7. (Color online) Magnetization curves for the non-SC sample BR 19 measured at different temperatures with the magnetic field applied along the c axis and within the ab plane. Data for 2 K (full squares) and 40 K (open squares) are almost coinciding.

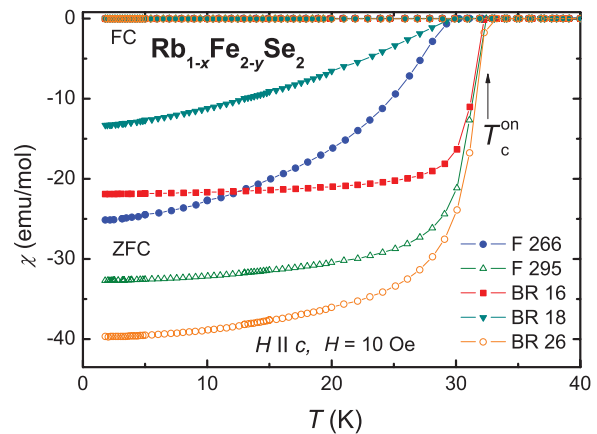


FIG. 8. (Color online) Temperature dependencies of ZFC and FC susceptibilities for different superconducting $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples measured in a field of 10 Oe applied along the c axis. The arrow indicates the temperature of the onset of the superconducting transition $T_c^{\text{on}} = 32.4$ K for sample BR 16.

susceptibilities for superconducting samples from different batches measured in a field $H = 10$ Oe applied parallel to the c axis. The sharpest transition ($\Delta T_c = 1.5$ K, estimated from FC data) and the highest temperature of the onset of superconductivity, $T_c^{\text{on}} = 32.4$ K was found for the samples from batches F 295, BR 16, and BR 26. Although the EPMA data presented above indicate some notable variations in the Fe and Rb concentrations for the samples from these batches, no significant broadening of the SC transition or variation of T_c^{on} was detected. At the same time, for the SC samples from the other batches (for example, F 266 and BR 18), the value of T_c^{on} was by ~ 2 K lower and the variation of χ_{ZFC} indicates a broader transition compare to the samples from batches F 295, BR 16, and BR 26. The reduction of the transition temperature and broadening of the transition for these samples probably can be attributed to disorder effects caused, for example, by larger deviations of the Rb and, especially, of the Fe concentrations from the “optimal” values of 0.8 and 1.6, respectively.

These “optimal” values of Rb and Fe concentrations correspond to a formal oxidation state for the Fe ions of +2.0 as found in other superconducting Fe chalcogenides. Within the experimentally determined ratio of Rb to Fe concentration (see Table I) we, however, did not found a change from the hole to electron doping with the variation of the stoichiometry. Indeed, for the samples with the lowest Fe concentration (below 1.5, e.g., BR17) the calculated valence of Fe is +2.27. For the samples with high Fe content (above 1.61, e.g., BR22), the valence of Fe is +2.07. Measurements of the thermoelectric power are necessary to clarify the role of doping on the properties of the Rb-based iron chalcogenides, as it was shown for the related K-intercalated compounds.^{42,43}

It is important to note that the values of χ_{FC} (Meissner effect) at 2 K for the SC samples are at least three orders of magnitude lower than the values of χ_{ZFC} (shielding effect), probably due to strong pinning. At the same time, the values of $4\pi\chi$ at 2 K calculated from the χ_{ZFC} data are far above unity, pointing to the dominance of demagnetizing effects. The measurements of χ_{ZFC} for the in-plane configuration (with H

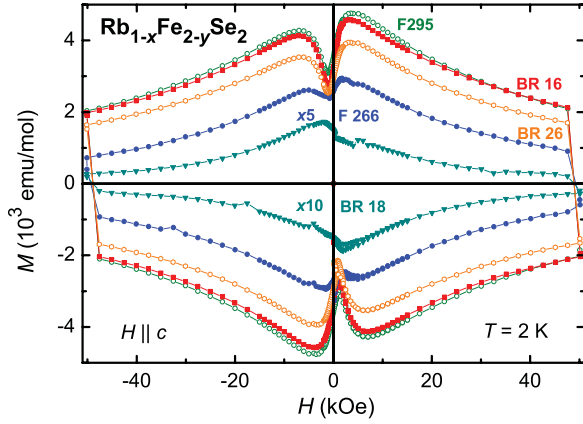


FIG. 9. (Color online) Hysteresis loops measured at 2 K with magnetic field applied along the c axis for different SC samples $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$. For clarity, the data for samples F 266 and BR 18 are magnified by factors of five and ten, respectively.

perpendicular to the c axis) with the negligible demagnetizing factor for the samples from batches F 295, BR 16, and BR 26 gave a value of $4\pi\chi$ close to unity that may indicate a large volume fraction of the SC phase, which yields a perfect 100% shielding.

Figure 9 presents the magnetization hysteresis loops for SC samples measured at 2 K with the magnetic field H applied along the c axis. The hysteresis loop for this configuration is symmetric with respect to the origin (0,0). The diamagnetic response for samples from different batches differs by a factor of 20. The reason of such a strong difference is unclear. It can be caused by variations of the stoichiometry detected by EPMA (see Table I) or by changes in the preparation conditions, such as oxygen contamination. The phase separation can also be critical for obtaining high SC hysteresis width. Indeed, the highest diamagnetic response was found in the samples from batches F 295, BR 16, and BR 26 in which x-ray diffraction revealed strong reflexes of the additional crystallographic phase. Additional pinning centers created by this additional crystallographic phase can enhance the diamagnetic response.

The critical current density j_c at 2 K estimated from the hysteresis data for the samples with the highest SC parameters using the Bean model for hard superconductors^{44,45} reaches a value of $1.6 \times 10^4 \text{ A/cm}^2$. It is by a factor of five lower than the intrinsic value of j_c reported for FeSe substituted with Te.^{46,47}

In Fig. 10, the hysteresis loop for the SC sample BR 16 measured at 2 K with the magnetic field applied perpendicular to the c axis is plotted. The maximum values of magnetization for $H \perp c$ and $H \parallel c$ correlate with the difference in the demagnetization factors for these two configurations. The hysteresis loop for the configuration $H \parallel c$ reveals two contributions, one superconducting, and the other linear in magnetic field. For comparison, in Fig. 10, the magnetization curve $M_{\parallel} = f(H)$ for the same sample measured at 40 K, just above the SC transition, is shown. It agrees well with the linear contribution that can be derived from the data at 2 K. This again points out that the magnetic correlations associated with the static antiferromagnetic order coexist with the SC state.

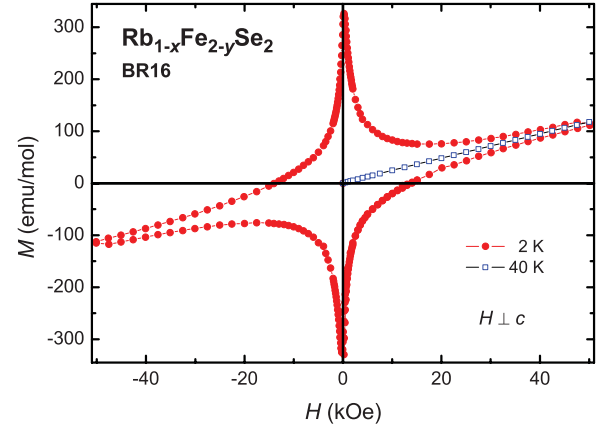


FIG. 10. (Color online) Hysteresis loop for SC sample BR 16 at 2 K (solid circles) measured with the magnetic field applied perpendicular to the c axis together with the magnetization curve at 40 K (open squares).

D. Resistivity

Figures 11(a) and 11(b) demonstrate the temperature dependence of the resistivity for the SC and non-SC samples, respectively. The resistivity measurements were done on rectangular samples by a four-point method in the temperature range 2–300 K using the resistivity measurement option of the Quantum Design physical properties measurement system (PPMS) in magnetic fields up to 90 kOe. Both, the in-plane resistivity, ρ_{ab} , with current j in the ab plane, and the interplane resistivity, ρ_c , with current j parallel to the c axis were measured. The magnetic field was always perpendicular to the current direction, so only the transverse configuration was studied. The contacts were made with a conductive silver paint. The error in the absolute value of the measured resistivity was about 20% due to finite dimensions of the potential contacts. The in-plane resistivity ρ_{ab} of the SC sample prepared by self-flux method (F 266) is by approximately one order of magnitude larger than that for samples grown by Bridgman method, which can be attributed to higher impurity content of the self-flux samples. The interplane resistivity ρ_c of a typical sample grown by Bridgman method with high SC parameters (BR 16) is by a factor of two higher than the in-plane resistivity. The resistivity curves of all superconducting samples show a broad hump at a temperature T_{max} that for different samples varies in the range of 190–215 K with a semiconductor-like temperature dependence above and a metal-like dependence below T_{max} . We note that the superconducting samples from batch BR 16 have the smallest value of the resistivity compared to other batches. It is also the smallest for the superconducting $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ crystals reported so far.^{12,15} Together with the larger resistivity ratio $\rho(T_{\text{max}})/\rho(T_c) = 37$, this may indicate a higher purity of these superconducting samples.

The resistivity of the nonsuperconducting samples is by several orders higher than that of the SC samples. It shows a steep increase on decreasing temperature and a small but clearly discernible anomaly at $T^* = 233 \text{ K}$. In the range 300–240 K, the temperature dependence of the resistivity for the non-SC samples can be described by Arrhenius law $\rho = \rho_0 \exp(\Delta E_a/kT)$ with an activation energy ΔE_a of 0.16, 0.079, and 0.073 eV for the samples BR 19,

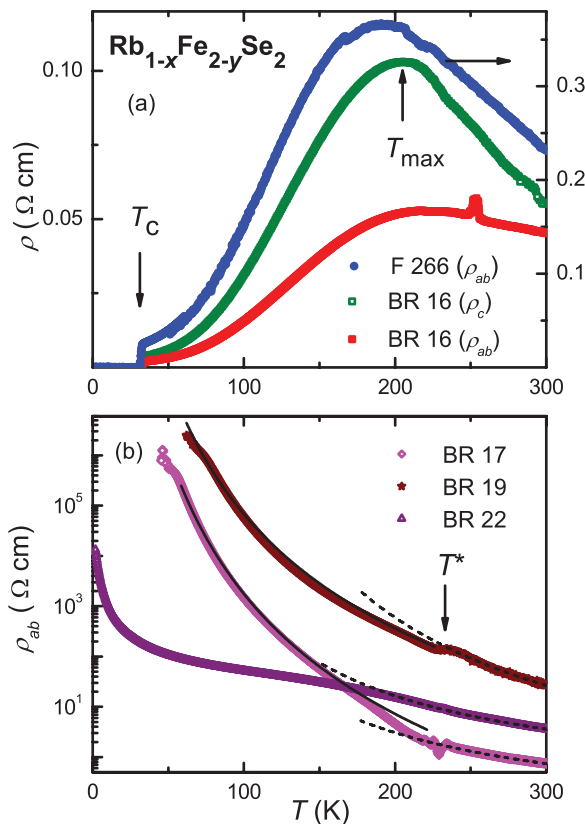


FIG. 11. (Color online) (a) Temperature dependencies of the in-plane resistivity ρ_{ab} and interplane resistivity ρ_c for the SC sample BR 16 (left scale) and for the in-plane resistivity for the SC sample F 266 (right scale) measured on cooling in zero external magnetic field. (b) Temperature dependencies of the in-plane resistivity for the non-SC samples. Dashed curves mark the Arrhenius fits at high temperatures, solid lines are the MVRH fits at lower temperatures as described in the text.

BR 22, and BR 17, respectively. At temperatures below 240 K down to 80 K, the resistivity of the non-SC samples is not thermally activated but, instead, can be reasonably described within a Mott variable-range hopping (MVRH) model by an expression $\rho = \rho_0 \exp(T_0/T)^{1/4}$. The resistivity of the sample BR 22 with the Fe concentration above 1.62 is also showing semiconductor-like behavior with decreasing temperature, but its value is by several orders of magnitude lower than that of the samples with the Fe concentration below 1.5. Summarizing the observed resistivity behavior of the nonsuperconducting samples, we notice rather low values of the activation energies, which are more typical for heavily doped semiconductors or Mott insulators than for intrinsic band insulators. Additional studies including Hall effect are necessary to clarify the actual transport mechanisms in the nonsuperconducting $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ crystals.

Figures 12(a) and 12(b), respectively, illustrate the effect of a transverse magnetic field on the in-plane ρ_{ab} and interplane ρ_c resistivities in the transition region for the SC sample BR 16. The measurements were done on warming after cooling in zero field. In zero field, the transition temperature, determined at the level of the 90% drop of the normal state resistivity ρ_n , coincides with the T_c^{on} (within 0.1 K) determined

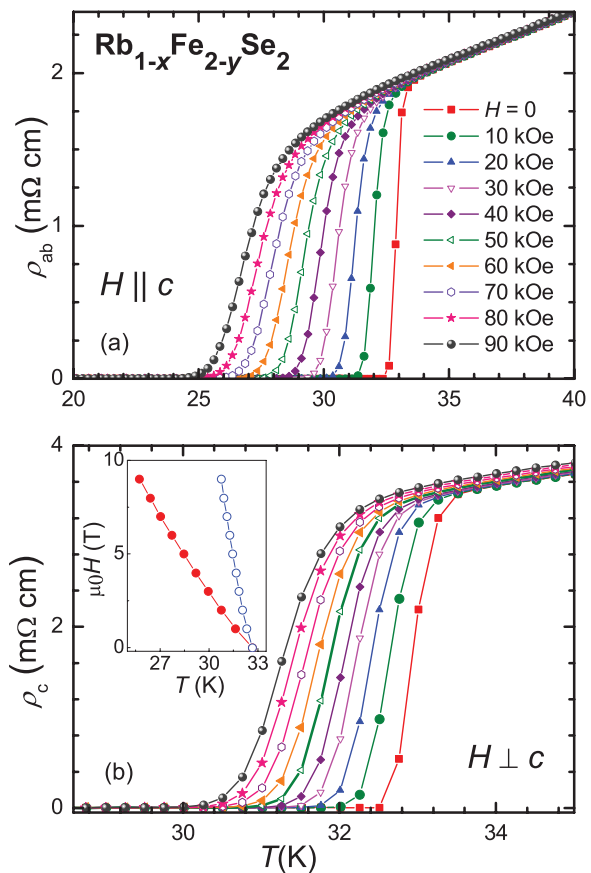


FIG. 12. (Color online) (a) Temperature dependence of the in-plane resistivity ρ_{ab} for the sample BR 16 measured in various magnetic fields applied parallel to the c axis. (b) Temperature dependence of the interplane resistivity ρ_c for the sample BR 16 measured in various magnetic fields applied perpendicular to the c axis. Inset: temperature dependence of the upper critical field for interplane (closed symbols) and in-plane (open symbols) configurations as described in the text.

from the susceptibility measurements. This again indicates the high quality of the superconducting samples. When increasing the magnetic field, the resistivity curves are displaced to lower temperatures. The inset in Fig. 12(b) shows the temperature dependence of the upper critical field $H_{c2}(T)$ estimated using the criterion of the 90% drop of the normal state resistivity. On approaching T_c , the slope of the $H_{c2}(T)$ curve becomes smaller compared to that at lower temperatures. The estimation of the upper critical field $H_{c2}(0)$ for $T = 0$ K made within the Werthamer-Helfand-Hohenberg model⁴⁸ using the relation $H_{c2}(0) = -0.69T_c(dH_{c2}(T)/dT)|_{T_c}$ gave a value of 250 kOe for the in-plane and 630 kOe for the interplane configurations. They are by a factor of two lower than those for FeSe substituted with Te^{46,47} and are the lowest reported so far for SC $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples.^{12,15}

E. Specific heat

Figure 13 presents the temperature dependence of the specific heat C for selected superconducting and nonsuperconducting samples. The heat capacity was measured by a relaxation method using the PPMS in the temperature range

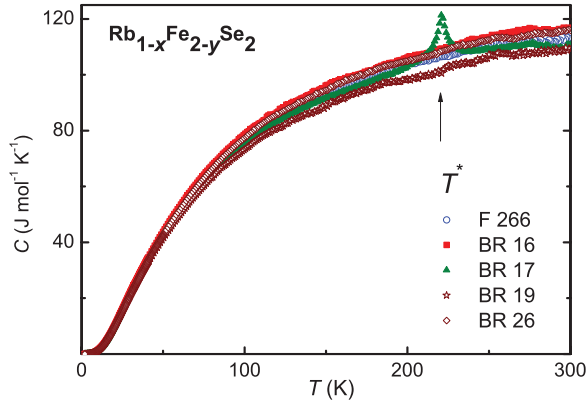


FIG. 13. (Color online) Temperature dependencies of the specific heat for different $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples. The arrow marks the anomaly in the specific heat at $T^* = 220$ K for the sample BR 17.

1.8–300 K and in magnetic fields up to 90 kOe. The magnetic field was applied parallel to the c axis of the samples. The specific heat for different samples exhibits very similar behavior and values in the measured temperature range. The specific heat of the non-SC sample BR 17 shows a sharp peak at around 220 K. Its position is lower than that of the peak in the susceptibility at 233 K for the same field configuration. Interestingly, the susceptibility of this sample for the field configuration perpendicular to the c axis exhibits an anomaly exactly at the same temperature of 220 K as observed in the specific heat. The application of the magnetic field of 50 kOe does not have any influence on the temperature position of the peak in the specific heat at 220 K indicating that it is not related to conventional AFM or FM transitions. Temperature-dependent x-ray measurements in the range 300–100 K did not find any evidence of a structural transformation either. Preliminary Mössbauer studies of the samples from this batch revealed new lines below T^* typical for trivalent Fe ions.⁴⁹ The observed anisotropy in the value of T^* for $H\parallel c$ and $H\perp c$ axes suggests a very peculiar type of this transition that needs further investigation.

In Fig. 14, the temperature dependence of the specific heat is shown in the representation C/T versus T^2 for different samples at temperatures below 12 K. A linear dependence with two different slopes below and above 5 K is clearly discernible. The values of the residual Sommerfeld coefficient γ_r were determined from the fit to the experimental data at temperatures below 3 K using the expression $C/T = \gamma_r + \beta T^2$,

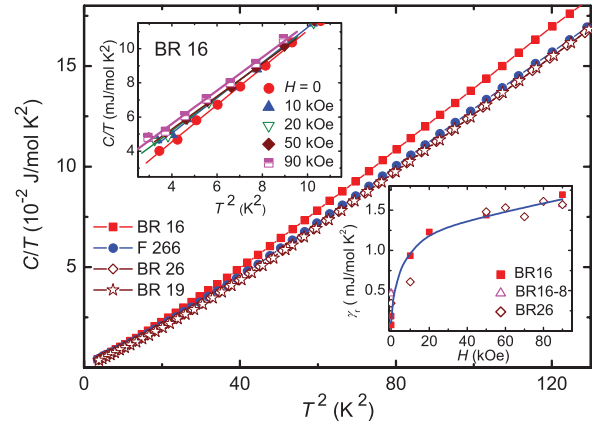


FIG. 14. (Color online) Temperature dependence of the specific heat in the representation C/T vs T^2 for different $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples at temperatures below 12 K. Upper inset: C/T vs T^2 in different applied magnetic fields for the sample BR 16. Lower inset: magnetic field dependence of the residual Sommerfeld coefficient γ_r for different superconducting samples with high SC parameters.

where the prefactor β characterizes both the lattice and magnon contributions to the specific heat, which are proportional to T^3 , and therefore cannot be separated. The calculated values of these parameters are given in Table II. We notice a rather low value of γ_r obtained for the superconducting samples, which indicates their high purity and minimal structural disorder or impurity content. The values of γ_r vary from 0.18 to 0.85 mJ/mole K^2 for the samples with high SC (BR 16) and poor SC parameters (F 266), respectively, reflecting higher impurity content and disorder of the poorly SC samples in agreement with the susceptibility results. Similar values of γ_r were reported for the related SC $\text{K}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ compound.⁵⁰ At the same time, the nonsuperconducting samples BR 17 and BR 22 exhibit rather high values of γ_r , which probably can be attributed to structural disorder due to Fe and Rb vacancies. Importantly, the nonsuperconducting sample BR 19 exhibits an extremely low value of γ_r indicating minimal disorder, being correlated with its best crystallinity revealed by x-ray studies (see Fig. 2). It should be noted that the value of β for the sample BR 19 is the closest to those of the superconducting samples. Together with very similar values of the susceptibility and its temperature dependence (see Fig. 5), this justifies the use of the specific heat data of the sample BR 19 as a reference for the lattice and magnon contributions. The respective contributions for the superconducting samples were calculated using the

TABLE II. Parameters determined from the susceptibility (T_c^{on} , ΔT_c) and heat capacity measurements.

Sample	T_c^{on} (K)	ΔT_c (K)	γ_r (mJ/mole K^2)	β (mJ/mole K^4)	γ_n (mJ/mole K^2)	$\Delta C/\gamma_n T_c$
F 266	30.9	2.8	0.85	0.96	10	1.0
BR 16	32.4	1.5	0.18	1.09	12	0.95
BR 16-8	32.4	1.6	0.46	1.04	12	0.96
BR 17	0.33	0.90
BR 19	0.95
BR 22	3.0	1.36
BR 26	32.4	1.5	0.34	0.94	7.4	2.2

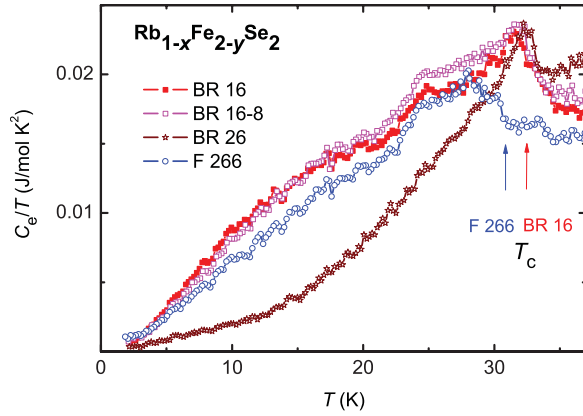


FIG. 15. (Color online) Temperature dependence of the electronic specific heat C_e/T vs T for different $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ samples calculated by subtraction of the magnon and lattice contributions. The arrows mark T_c for the samples F 266 and BR 16.

specific heat data of the sample BR 19 scaled by the square root of the ratio of their molar mass.

Figure 15 shows the electronic specific heat in the representation C_e/T versus T for several superconducting samples from different batches obtained by subtraction of the lattice and magnon contributions from the total measured specific heat. It has been found that the superconducting contribution to the specific heat is rather small compared to that of the lattice and magnon contributions. However, a pronounced lambdalike anomaly of C_e at the superconducting transition is clearly evidenced for all SC samples. The sample F 266 with poor SC parameters exhibits a broad anomaly of C_e at T_c , which is displaced to lower temperatures compared to the anomaly in samples from the batch BR 16 with high SC parameters. The samples from batch BR 26 show the sharpest anomaly of C_e at T_c .

The value of the Sommerfeld coefficient in the normal state γ_n for the SC samples was estimated from the constraint of entropy conservation at the onset of T_c , i.e., $\int_0^{T_c} C_e/T dT = \int_0^{T_c} \gamma_n dT$. The values of γ_n are given in Table II. They vary between 7.4 and 12 mJ/mole K² for SC samples from different batches and are comparable to those reported for SC $\text{K}_{1-x}\text{Fe}_{2-y}\text{Se}_2$.⁵⁰ Assuming that the residual Sommerfeld coefficient γ_r corresponds to the fraction of the normal conducting state, the obtained ratio of γ_r/γ_n implies that the volume fraction of the superconducting phase in our samples varies within 92–98%. The values of the jump of the specific heat $\Delta C/T_c$ at T_c were calculated by subtracting the value of γ_n from the maximum value of C_e/T at T_c . They vary from 10 to 16 mJ/mole K² for different SC samples and are by a factor of ~ 5 smaller than the respective values of $\Delta C/T_c$ observed for single crystalline $\text{FeSe}_{0.5}\text{Te}_{0.5}$.^{47,51,52} The reduced specific heat jump at T_c , $\Delta C/\gamma_n T_c$ varies from 0.95 (for BR 16) to 2.25 (for BR 26). The values below 1.0 are probably related to errors in the estimation of γ_n for these samples. The obtained value of $\Delta C/\gamma_n T_c = 2.25$ for the sample BR 26 with the sharpest anomaly in the specific heat at T_c is higher than the BCS estimate of 1.43 for the weak-coupling limit and may indicate strong-coupling superconductivity in $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ similar

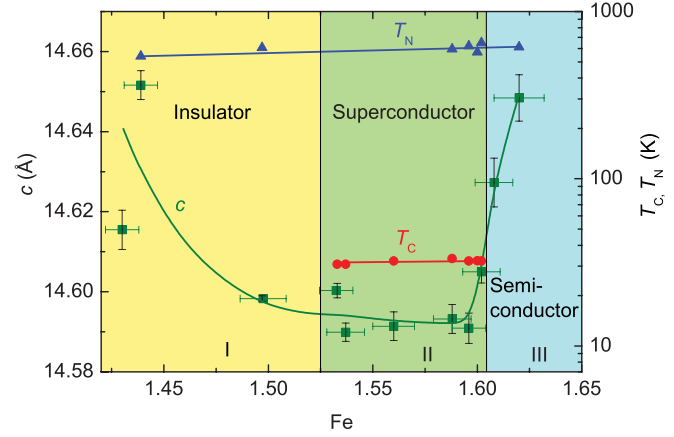


FIG. 16. (Color online) Phase diagram of $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ system presenting dependencies of the lattice constant c , transition temperature T_c , and the Néel temperature T_N for three different regions with distinct structural, magnetic, and conducting behaviors.

to that observed in Te substituted FeSe ⁵² and in optimally doped $\text{Ba}_{1-x}\text{K}_x\text{Fe}_2\text{As}_2$.^{53–55}

Since specific heat in the SC regime varies substantially among samples with rather close stoichiometry, we did not attempt to estimate the value of the superconducting gap. However, to get an insight into the symmetry of the order parameter, we measured the low-temperature specific heat in different magnetic fields. The upper inset in Fig. 14, where the specific heat in the representation C/T versus T^2 is plotted for one of the samples with high SC parameters (BR 16), demonstrates the effect of the magnetic field on the specific heat at lowest temperatures. The application of the magnetic field results in an increase of the residual Sommerfeld coefficient γ_r probably related to the pair-breaking effect. The field dependence of γ_r for three samples with high SC parameters are shown in the lower inset in Fig. 14. The values of γ_r for these samples are very close and exhibit a similar field dependence that significantly differs from the linear dependence expected for fully gapped superconductors. It must be noted that the dependence of $\gamma_r(H)$ is close to \sqrt{H} characteristic for d -wave symmetry.⁵⁶ Recently, a Volovik-like \sqrt{H} term in the specific heat was reported for an optimally doped $\text{BaFe}_2(\text{As}_{0.7}\text{P}_{0.3})_2$ pnictide, which indicate the presence of nodes in the superconducting gap.⁵⁷ However, there is no experimental evidence of nodes in $\text{A}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ yet. On the contrary, very recent ARPES studies of related $\text{A}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ materials^{22,23} seem to fully rule out d -wave pairing. At present, the sublinear dependence of the $\gamma_r(H)$ observed in $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ is lacking any reasonable theoretical explanation but the multiband effect with two different gaps discussed in Ref. 57 can be also relevant in this case.

III. SUMMARY AND CONCLUSIONS

Our detailed preparative, structural, magnetic, conductivity, and specific heat studies of $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ single crystals testify several important peculiarities of these materials. (1) Three different regions with distinct structural, magnetic, and conducting behavior of the samples are documented in the $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ system as presented in the phase

diagram in Fig. 16. Superconductivity is observed in the range of the Fe concentration above 1.53 and below 1.60 (region II of the diagram). Superconducting samples with the sharpest transitions, the highest critical temperature of 32.4 K and the largest diamagnetic response have the “optimal” composition $\text{Rb}_{0.8}\text{Fe}_{1.6}\text{Se}_2$. The reduction of the superconducting parameters of the samples can be attributed to deviations of the Fe concentration from the “optimal” value of 1.6. Within the experimentally determined range of the Rb concentration (0.6–0.8), no correlation between the variation of the Rb content and the lattice parameters of the samples was found. The superconducting samples show the smallest value of the lattice parameter c compared to the nonsuperconducting samples. Samples with the Fe content below 1.5 (region I) and above 1.6 (region III) show an insulating, or semiconducting behavior at low temperatures, respectively. (2) In the whole range of the studied Fe concentrations, the susceptibility measurements evidenced a behavior typical for an anisotropic antiferromagnet with a transition temperature T_N above 500 K. (3) Magnetization measurements provide an evidence of the presence of antiferromagnetic correlations in the ordered superconducting state in samples with the Fe concentration in the range of 1.53–1.6. This suggests the coexistence of superconductivity and static antiferromagnetic order. In view of recent experiments,^{38–40} this coexistence should be considered as being on the macroscopic level within a scenario of the phase separation. (4) Magnetic hysteresis measurements of the superconducting samples revealed high values of the critical current density j_c of 1.6×10^4 A/cm² for superconducting samples with the composition close to $\text{Rb}_2\text{Fe}_4\text{Se}_5$. The obtained value of j_c is the largest reported

thus far for $\text{A}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ materials. (5) The upper critical fields H_{c2} of ~ 250 kOe for the in-plane, and 633 kOe for the interplane configurations are estimated from the resistivity study of the superconducting samples in magnetic fields. They are the lowest reported to date for the $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ superconductors. (6) Specific-heat measurements of the superconducting samples evidenced very low values of the residual Sommerfeld coefficient corresponding to 92–98% volume fraction of the superconducting phase, which confirms the bulk nature of the superconducting state. (7) Resistivity studies revealed an insulating behavior at low temperatures for samples with Fe concentrations below 1.5. The conductivity of these samples at low temperatures can be reasonably described by the variable-range hopping Mott mechanism. The conductivity of the samples with Fe concentrations above 1.6 is typical for heavily doped semiconductors. (8) In the nonsuperconducting samples with the Fe concentration below 1.45, an unusual phase transition below 233 K is revealed, which is not related to antiferromagnetic or structural transformations. (9) The discrepancies in the range of Fe concentrations, where superconductivity shows up in $\text{Rb}_{1-x}\text{Fe}_{2-y}\text{Se}_2$ as compared to earlier reports, are likely to be related to the accuracy of the methods used for the compositional analysis.

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